L. Małdziński, J. Tacikowski

Concept of an economical and ecological process of gas nitriding of steel

An economical, ecological version of the gas nitriding process is presented. It allows a substantial reduction in the consumption of industrial gases compared with currently used processes. Also it leads to a simplification of nitriding installation as well as the process itself, while allowing maintenance of full control over the kinetics of the layer growth. The concept was developed from studies of the thermodynamics and kinetics of growth of nitrided layers on iron and steel. It has been verified in trials under industrial conditions.

Es wird eine ökonomische, ökologische Möglichkeit des Nitrierprozesses mit Gas vorgestellt.

1 Introduction

The nitriding process, which improves the life expectancy of machine parts, may reduce at the same time the consumption of steel and energy and, as a result, the cost of the entire manufacturing process. For this reason, it may be qualified as economical and ecological.

Economical and ecological advantages will at the same time characterize a process, in which the formation of the nitrided layer occurs in the shortest time, with minimal consumption of gases, electrical energy, and water. Concurrently, it is assumed that the nitrided layer has an optimal structure in terms of expected performance i. e. the required zones of iron nitrides $(\alpha, \gamma' + \alpha \text{ or } \varepsilon + \gamma' + \alpha)^1$, correct thickness, nitrogen concentration profile and hardness, and a surface condition that allows immediate use of parts without any finishing operations.

Ecologically, the amount and composition of gas emissions emitted into the surrounding atmosphere are also important. Exhaust gas should contain only environmentally neutral components such as H_2 , N_2 and H_2O and be free of nitrogen oxides or residual NH₃. Only by satisfying all these requirements the process is considered to be both, economical and environmentfriendly, and therefore ecological.

The present concept of an economical, and at the same time ecological gas nitriding process creates the conditions for a substantial reduction in the consumption of industrial gases, as compared with currently used processes. Also, it leads to a simplification of the nitriding installation as well as of the process itself, while allowing full control over the kinetics of the layer growth. The process, based on the thermodynamics and kinetics of growth of nitrided layers, has been tested and verified in industrial application.

2 Review of current processes of gas nitriding

To date, traditional nitriding, with very limited control over the growth of the nitrided layer is still very much in use worldwide [1]. It is performed using only ammonia and the resulting layer is composed of superimposed $\varepsilon + \gamma' + \alpha$ phases. In industrial practice, the superficial zone of iron nitrides $\varepsilon + \gamma'$ (as a rule exceedingly thick and brittle) is usually removed by grinding [2, 3], which substantially adds to the cost of the manufacturing process. Moreover, as will be demonstrated later, the formation of an ϵ + γ' + α layer uses much more ammonia (NH₃) than the formation of a layer composed of $\gamma' + \alpha$. It is because the growth of an $\varepsilon + \gamma' + \alpha$ layer occurs at a much higher nitriding potential (K_N) value. This then calls for a greater flow of NH3 through the retort and consequently its higher consumption. The lowest potential K_N and the lowest NH₃ consumption is required for the creation of the diffusion (precipitation) layer α . Therefore, in the traditional production process, the $\varepsilon + \gamma'$ layer and its subsequent removal is, from the economical point of view doubly disadvantageous.

Controlled gas nitriding, performed for over 60 years, represents an advancement over an earlier (but still encountered today) 'traditional' method, using 100 % ammonia atmospheres. Two-component atmospheres of ammonia and dissociated ammonia ($NH_3 + NH_3$ diss.) [4, 5], as well as ammonia and molecular nitrogen $(NH_3 + N_2)$ came into use about 50-60 years ago [6-8]. Through proper selection of atmosphere composition and adjustment of atmosphere flow rate through the retort, it is possible to form layers of the required phase composition (consisting of $\varepsilon + \gamma' + \alpha$, $\gamma' + \alpha$ or α zones) of required thickness of individual zones, along with a limited nitrogen concentration at the surface. Diluting ammonia with dissociated ammonia $(NH_3 diss.)$ or with molecular nitrogen (N_2) results in a diminishing supply of atomic nitrogen to the surface, leading to a reduction of the superficial nitrogen concentration, and in consequence to a limitation of growth or even to a removal of the undesirable brittle nitride zones. Also, similar layers with a limited ϵ or $\epsilon + \gamma'$ zone can be obtained in the 'vacuum nitriding' process (Nitrovac) [9], or through plasma nitriding [2, 3].

These processes allow the elimination of final grinding, thus reducing the cost of the whole manufacturing process. However, as will be shown below, nitriding using two-component atmospheres involves a considerable consumption of gases, hence a higher cost of the process.

From the two mentioned atmosphere types, $NH_3 + NH_3$ diss. and $NH_3 + N_2$, the second one (with N_2) is more widely used in industrial practice. It is usually cheaper than the $NH_3 +$ NH_3 diss. atmosphere, as nitrogen N_2 is in most cases cheaper than dissociated ammonia. Moreover, production of NH_3 diss. requires a dissociator, which adds to the cost of the installation. The dissociator, generating a high-hydrogen mix often represents a psychological barrier to potential industrial users. On the other hand, in the case of $NH_3 + N_2$ atmospheres, the control of the kinetics of layer growth is much more difficult and limited because of the non-equilibrium character of the process [10]. The same is true for plasma nitriding [2, 3].

¹ The term $\varepsilon + \gamma' + \alpha$ represents a nitrided layer composed of: ε at the surface, γ' more or less beneath ε and an α diffusion zone underneath; the term $\gamma' + \alpha$ denotes a layer composed of γ' at the surface and an α diffusion zone underneath; the term α denotes an α diffusion zone adjacent to the surface.

Installations of vacuum and plasma nitriding are technically more complex and therefore more expensive than installations of controlled nitriding [3, 4].

3 Essentials of the concept of economical and controlled nitriding

The present concept assumes carrying out the process of controlled nitriding with NH₂ alone. This is a simpler process than that using two-component mixes of ammonia diluted with N₂ or NH₃diss.² The regulation of the chemical composition of the atmosphere in the retort, and therefore the regulation of $K_{\rm N}$, is obtained through the variation of the inflow rate of NH₂ into the retort³. The authors accept that the inflow of NH₃ into the furnace may at times be reduced to zero, which widens the range of the gas composition in the retort (and by the same token, of the nitriding potential K_N). The amount of NH_3 introduced into the furnace is regulated (controlled) with the aid of a gas analyzer (for example an infrared analyzer). The gas analyzer opens or closes periodically the NH₃ inlet valve at a frequency required for maintaining a desired gas composition (a desired K_N) in the retort. The concept presented here is not an entirely new solution. In existing practice, from early years on, the composition of the atmosphere has been controlled through the variation of the flow rate of NH₃ into the furnace, but in a relatively narrow range, without closing the inflow entirely. In rare instances (aerospace industry), considerable limitations of ammonia inflow have been applied, accompanied by a large drop in the K_N potential. However, in the case of the most widely used processes of controlled nitriding, the established practice is focused on "gas-guzzling" processes using two-component diluted atmospheres. This raises the question: why, to this day, has gas nitriding based on a single-component – NH₃ – atmosphere has not been implemented on a larger scale?

It may be supposed that the reasons have been: an insufficient knowledge of the fundamental physico-chemical mechanisms of nitriding as well as the complexity of nitriding processes in industrial installations. Literature on the subject does not answer the question whether the dissociated ammonia formed in the furnace (on the surfaces of the retort and of the load) is equivalent in its effect on the nitriding kinetics to $\rm NH_3 diss$. generated in an upstream dissociator. Therefore there was no answer to the question whether a mix containing $\rm NH_3 + NH_3 diss$. formed in the retort would lead to the same surface nitrogen concentration, and as a consequence to the same layer structure as a $\rm NH_3 + NH_3 diss$. mix with $\rm NH_3 diss$. supplied from a dissociator, assuming the same nitriding potential $\rm K_N$ in both cases.

There was an unfounded opinion among many practitioners that the rate of the inflow of ammonia (and also of the $\rm NH_3 + N_2$ and $\rm NH_3 + NH_3$ diss. mixes) into the retort has an effect not only on the atmosphere composition in the furnace but also, independently, on the kinetics of the layer growth. They considered, apart from maintaining a required nitriding potential $\rm K_N$, for ensuring a correct growth kinetics, it also necessary to provide a certain flow rate of the atmosphere through the retort. For this reason, relatively high flow rates of atmosphere have been used to this day.

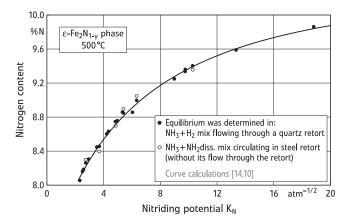


Fig. 1. Absorption isotherm of nitrogen in the ε -Fe₂N_y phase, determined experimentally in atmospheres obtained from NH₃, NH₃ + H₂ and NH₃ + NH₃diss. Curve – calculated according to [14, 10]

In the following section, we discuss the essential aspects of the proposed concept of controlled nitriding, employing $\rm NH_3$ only. These aspects will involve thermodynamic, kinetic and technological considerations. In the final section the economical impact will be considered.

2.1 Thermodynamic aspect of nitriding in atmospheres obtained from NH_3 and from two-component $NH_3 + H_2$ and $NH_3 + NH_3$ diss. mixes

Małdziński et al. [11-14] investigated the equilibrium between the atmosphere obtained from NH₃, or NH₃ + H₂, NH₃ + NH₃diss. and NH₃ + N₂ mixes, and nitrogen in the γ' and ε phases of Fe-N alloys. They used thin, porous (open-pore structure) iron foils. Processing was performed in a laboratory furnace with a quartz tube and also in an industrial furnace with a steel retort and a circulating fan. The main difference between the two experiments was that in the quartz retort there was practically no dissociation of NH₃ while it was considerable in the steel retort. This means that in the steel retort a substantial amount of NH₃diss. was generated on the walls of the retort and surfaces of the charge racking.

These studies demonstrated that the equilibrium conditions between the retort atmosphere and nitrogen in the nitride phases are neither affected by the type of the atmosphere nor that of the furnace, but only by the composition of the atmosphere or, in other words, the nitriding potential K_N at the sample surface. The absorption isotherm of nitrogen dissolved in ϵ -Fe₂N_{1,9}⁴, shown in Fig. 1, has been determined experimentally, taking into account data points obtained for: NH₃, the NH₃ + H₂ and NH₃ + NH₃diss. mixes and the two furnace types, one with quartz- the other with the steel retort and fan. In the quartz retort, the equilibrium was obtained in atmospheres NH₃ + H₂ and NH₃ + NH₃diss. in conditions of their constant flow (filled circles). In the furnace with the steel retort, the equilibrium was achieved using only NH₃, with its supply entirely closed, though with an operating circulation fan (open circles).

The above results confirm that the equilibrium between the atmosphere in the retort and nitrogen in the Fe-N phases depends only on the composition of the atmosphere, or K_N , at the nitriding surface.

Also, when the $\rm NH_3$ supply to the retort is entirely stopped (in the furnace with the steel retort), the equilibrium concentration of nitrogen in the γ' and ϵ phases is a function of the K_N poten-

 $^{^2}$ This refers to $\rm N_2$ and $\rm NH_3 diss.$ introduced together with ammonia into the retort, and not to the same components formed in the retort as products of $\rm NH_2$ dissociation.

¹³ The inflow rate of atmosphere into the retort is more relevant to this discussion than the flow rate *through* the retort. The latter, because of NH_3 dissociation and thermal expansion of gases, is difficult to measure and practically unimportant.

⁴ The absorption isotherm of nitrogen dissolved in ϵ -Fe₂N_{1-y} denotes the relationship between the nitriding atmosphere (nitriding potential K_N) and the amount of nitrogen in the ϵ phase under equilibrium conditions at constant temperature.

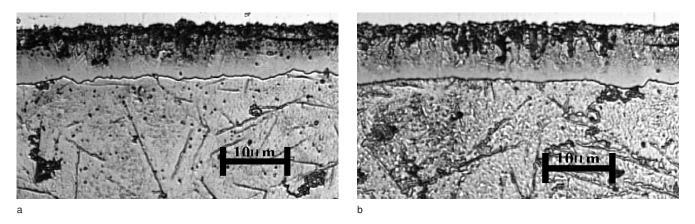


Fig. 2. Micrographs of the nitrided layer on Armco iron, nitrided in a furnace with a quartz retort at 550 °C, $K_N = 1.3 \text{ atm}^{-1/2}$, for 10 hours, with a linear velocity along the sample surface: (a) 1 cm/s, (b) 7.5 m/s

tial only. In these experiments different linear atmosphere flow velocities along the sample surface were used. In the quartz retort this velocity varied between 3 cm/s and 10 cm/s, in the steel retort, due to the action of the fan, varied from 0,2 m/s to 1,5 m/s. This result indicates that the linear velocity of flow along a sample surface does not affect the equilibrium between the nitriding atmosphere and nitrogen in the iron foil.

On the basis of the above considerations it follows that from the perspective of thermodynamics, dissociated ammonia generated in a separate dissociator is equivalent to the same NH_3 diss. formed inside the steel retort (furnace with a fan).

3.2 Kinetic aspect of nitriding in atmospheres obtained from NH_3 and from two-component $NH_3 + H_2$ and $NH_3 + NH_3$ diss. mixes

The above conclusion is also important from the point of view of kinetics, as the nitrogen concentration at the surface determines the growth rate of the layer and its phase structure. But in the condition of equilibrium between the atmosphere and nitrogen in a thin iron foil there is no net diffusion of nitrogen atoms from the surface into the sample core. This raises the question of the influence of the above-analyzed factors on the surface concentration of nitrogen $(c_{\scriptscriptstyle N}{}^s)$ and, in consequence, on the kinetics of layer growth on samples of considerable thickness. In this event there is a diffusion of nitrogen atoms from the surface into the core, and the surface concentration of nitrogen tends be determined by the relative fluxes of nitrogen atoms from the atmosphere to the metal surface and from the surface into the metal [15-18]. In such a case, the linear component of atmosphere flow velocity parallel to the metal surface can also have an effect on the layer growth. If it is too small, it may slow down the growth kinetics. On the other hand, some practitioners are concerned that too high a linear velocity may also have a detrimental effect on the kinetics of nitriding. It could supposedly 'destroy' the Nernst layer and as a result impede the supply of atomic nitrogen to the surface.

Within the scope of this investigation, the effect of linear flow velocity of the $\rm NH_3 + NH_3$ diss. atmosphere along the sample surface on the kinetics of growth of the $\epsilon + \gamma'$ zone on Armco iron was studied. The experiments were carried out using a quartz tube furnace, within the range of linear flow velocity from 0.25 cm/s to 7.5 m/s, at a constant temperature and nitriding potential $\rm K_N$. It turned out that the flow velocity of atmosphere parallel to the nitrided surface has only an effect with very low velocities 0.25–1 cm/s, and this solely in the initial period of the

nitriding process, when the absorption of nitrogen by the iron surface is highest⁵. In the range of 1 cm/s to 7.5 m/s the linear velocity has no effect on the growth of the above-mentioned layer anymore. The $\varepsilon + \gamma'$ zone on Armco iron shown in Fig. 2a was obtained with a linear velocity of 1 cm/s, and with 7.5 m/s in Fig. 2b. Both cases yielded a similar thickness of the whole $\varepsilon + \gamma'$ layer and of the individual ε and γ' zones. These results show conclusively that the kinetics of layer growth does not depend on the linear flow along the treated surface but only on the composition of the atmosphere.

In the presence of nitrogen diffusion into the metal, the type of the employed two-component atmosphere is not unimportant. In the $\rm NH_3$ + $\rm N_2$ atmosphere with a high $\rm N_2$ content, the supply of the atomic nitrogen to the surface is reduced, thus decreasing the nitrogen concentration at the surface. In this event the process is controlled by two factors: the nitriding potential $\rm K_N$ of the atmosphere in the retort, and the degree of its dilution with $\rm N_2$. In an atmosphere obtained from $\rm NH_3$ or $\rm NH_3$ + $\rm NH_3$ diss. $\rm N_2$ is also present, but its content is low and in practice the process is controlled only by the potential $\rm K_N$.

3.3 The technological aspect of nitriding in industrial furnaces

In a typical furnace, incorporating a steel retort and a circulating fan, the atmosphere composition depends directly on three basic factors [10, 19]:

- the dissociation rate of NH3 in the furnace retort;
- circulation flow of the nitriding atmosphere in the retort;
- inflow of nitriding atmosphere (NH₃, NH₃ + NH₃diss., NH₃ + N₂) into the retort.

Dissociation of $\rm NH_3$ results in the drop of potential $\rm K_{N'}$, and thus in a decreased supply of nitrogen to the metal surface. Insufficient circulation of atmosphere in the retort may lead to an inhomogeneity of the atmosphere composition.

The circulation (imposed by a fan) produces a high linear velocity of the atmosphere along the surface of the nitrided charge. It improves the homogeneity of the atmosphere composition in the retort [10, 19]. It also improves the uniformity of temperature. In addition, in the early stages of the process, circulation increases the heat flow from the retort walls to the charge.

⁵ This velocity is so low that, were it not for a risk of atmosphere inhomogeneity, no fan would be necessary and convection flows of gas in the retort would be sufficient. A linear velocity of the order of 7.5 m/s may be found in modern industrial furnaces.

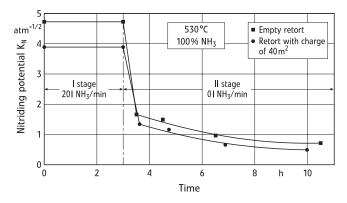


Fig. 3. Decrease in the nitriding potential K_N after closing NH_3 inflow into the retort, in an empty furnace (retort surface area – 11 m²) and in the same retort with charge (surface area – 40 m²)

In turn, a higher inflow rate of NH_3 (or of a two-component atmosphere) counteracts the drop of K_N and, in a certain range of flow and dissociation, rates may improve or worsen the homogeneity of atmosphere in the furnace [10, 19].

4 Experimental verification of the concept in the case of an industrial furnace

Within this investigation two experiments with two-stage processes were carried out. They both involved, in the initial period of the process (3 h), a high ammonia inflow into the retort, followed by closing the inflow entirely.

The first of these experiments was carried out with an empty retort (nominally 11 m² surface area of the retort, no charge), the second with a large-surface charge (ca. 40 m² surface area). Fig. 3 shows the variation of the nitriding potential as a function of time, for both processes. It illustrates that after stopping the supply of $\rm NH_3$ to the retort, the potential K_N falls rapidly at first, then very gradually. This pattern appears regardless of whether the retort is empty or filled, and the differences in K_N between the two processes are not too big.

The rapid initial drop of K_N is a favourable phenomenon from the point of view of process regulation and control. It allows a low level of potential to be reached quickly, where further regulation of composition may be effected, through a continuous or pulsating dosage of NH₃ to the retort⁶.

The comparison of the two curves of the variation of K_N also indicates that the dissociation rate of NH_3 is mainly determined by the surface area of the retort. The charge, despite its much larger surface area, influences NH_3 dissociation to a lesser degree. This happens for two reasons. Firstly, the increase in the catalytic surface of the retort and charge does not result in a proportional increase of the dissociation rate of NH_3 because as the ammonia content in the atmosphere decreases (due to an increase in the surface area of the retort and charge), the driving force of the dissociation reaction is diminished. Secondly, the effective catalytic surface area of the retort is, after a longer service, much larger than the surface of the charge. The influence of the surface area on the dissociation is therefore greater than that of the charge.

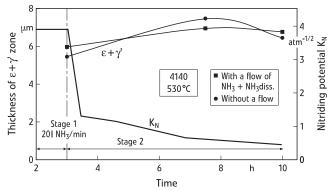


Fig. 4. Kinetics of growth and reduction of the ε + γ' zone in a two-stage process: stage 1: 100 % NH₂, stage 2: NH₃ inflow entirely closed (variant 1) or atmosphere changed to NH₃ + NH₃diss. (variant 2)

In the following section of the investigation a comparison was made between the growth of the e+g zone on 4140 steel during nitriding according to the new concept (with occasional stopping of the ammonia flow into the retort) and the growth of the same zone in a controlled standard process. In both cases a two-stage process was used. Stage 1, the same in both compared processes, was carried out in 100 % NH₃, at $K_N = 4$ atm^{-1/2} for three hours. Stage 1, according to the new concept, was run with a closed NH₃ supply. In the course of the process the potential decreased, as plotted in the graph in Fig. 3. In variant 2 of the process, stage 2 was run using the NH₃ + NH₃diss. atmosphere, the composition of which was varied so as to produce the same time-dependent decrease of the potential K_N as in the first variant.

Fig. 4 also shows the changes in the thickness of the $\varepsilon + \gamma'$ zone, formed in both processes. As may be seen, differences in the thickness of comparable zones are small and practically negligible. These results prove again that closing the NH₃ supply to the retort does not affect the growth rate of the nitrided layer. Growth is determined only by the process parameters (T, K_N and time). In the following two processes, samples of the 4140 steel were inserted into the retort during stage 2. Fig. 5 shows the surface hardness (measured with the Vickers

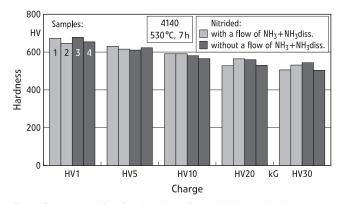


Fig. 5. Comparison of surface hardness of nitrided 4140 steel, obtained in a stage 2 (as in Fig. 3) with $NH_3 + NH_3$ diss. inflow, and in a process with a closed supply of 100 % NH_3 . Variation of potential during second stage as in Fig. 3. Time – 7 h

 $^{^{\}overline{6}}$ During the drop of K_N , the possibility of regulation (or control) of the growth of the nitrided layer is suspended until the potential reaches a level programmed for the 2^{nd} process stage. From this moment onwards K_N is regulated through a continuous, or intermittent inflow of NH_3 into the retort.

pyramid under loads of 1 kg to 30 kg). The lack of difference between the hardness levels of both layers is evidence of the same kinetics of nitriding, whether the $\rm NH_3$ supply to the retort is constant or occasionally closed.

The experiment was continued for a total duration of 10 hours. Its extension would, in accordance with the laws of thermodynamics, cause almost complete dissociation of $\rm NH_3^{-7}$. Thus, prolonging the process would require a small, constant supply of $\rm NH_3$ in order to maintain the required nitriding potential in the retort.

5 Comparison of consumption of NH_3 , $NH_3 + NH_3$ diss. and $NH_3 + N_2$ atmospheres

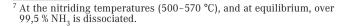
For a comparison of consumption of $\rm NH_3$, $\rm NH_3 + \rm NH_3 diss.$ and $\rm NH_3 + N_2$ atmospheres, it has to be assumed that processes using these gas mixes are carried out at the same temperature, and the resulting nitrided layers have the same phase structure, same thickness and the same concentration profiles of nitrogen in individual zones. In other words, that in the analyzed processes the same amount of nitrogen is introduced into the metal in every case.

This is equivalent for each of the atmospheres under consideration, to the equality of diffusion flows of nitrogen, J_N^g , into the metal. Satisfying this condition requires setting up a specific composition of the atmosphere in the retort. This composition, in the case of 100 % NH₃, is determined only by the inflow rate, while for NH₃ + NH₃diss. and NH₃ + N₂, by both the inflow and the ratio of their components.

5.1 Comparison of consumption of NH_3 and $NH_3 + NH_3$ diss. atmospheres

A comparison of consumption of NH₃ with that of NH₃ + NH₂diss. is relatively simple. In order to ensure the equality of flows of nitrogen atoms from either of the atmospheres to the surface, it is necessary to adjust the intensity of flow of NH₃ and of NH₂ + NH₂diss. and the composition of the latter so as to obtain in the retort, in both cases, the same nitriding potential K_{N} . According to previous considerations (sections. 3.1, 3.2 and 3.3), in processes using NH_3 and NH_3 + NH_3 diss. the nitriding potential K_N is the only factor determining the flux of nitrogen atoms J_N^{g} from the gas phase to the surface and, in consequence, the kinetics of the layer growth. In order to analyze the consumption of atmosphere in both cases we utilized the relationships, determined experimentally on an industrial furnace, between the flow of the atmospheres NH₃ and NH₃ + NH₃diss., and the nitriding potential K_N inside the retort. Such relationships are shown in Fig. 6 for 530 °C.

It illustrates that in order to replace 100 % NH₃ with a 60 % NH₃ + 40 % NH₃ diss. mix., maintaining the same potential $K_N = 0.35 \text{ atm}^{-1/2}$, it is necessary to increase its flow 7.7 times, in comparison with 100 % NH₃. Taking into account the expansion of NH₃ diss, this is equivalent to a 5.6-times higher consumption of NH₃. Still larger differences would appear in the case of replacing the NH₃ with a 20 % NH₃ + 80 % NH₃ diss. mix (this mix is frequently used in the stage 2 of nitriding of alloy steels). In this case (Fig. 6), in order to maintain the potential of 0.35 atm^{-1/2}, the flow of 20 % NH₃ + 80 % NH₃ diss. would have to be increased 50 times, compared with the inflow of pure NH₃. This is equivalent to a 30 times higher consumption of NH₃ when using a two-component mix. From this comparison, it also emerges that diluting NH₃ with dissociated NH₃ diss. leads to an increased consumption of NH₃, and the higher the dilution,



2.0_Г 1/2 atm 86% NH3+14%NH3diss. 100% NH3 ¥ 1.5 Nitriding potential 60% NH3+40%NH3diss. 1.0 Commercial furnace 530°C The flow increase 0.5 20% NH3+80%NH3diss. 0 4 l/min 0 2 6 8 10 The inflow rate of NH₃ and NH₃+NH₃diss.

Fig. 6. Effect of the inflow rate of NH_3 and NH_3 + NH_3 diss. atmospheres on the nitriding potential in a furnace retort. Industrial nitriding pit-type furnace, 1 m³ working chamber

the higher the consumption. Replacing 100 % $\rm NH_3$ with a $\rm NH_3$ + $\rm NH_3 diss.$ mix at other temperatures shows the same tendency.

5.2 Comparison of consumption of $\rm NH_3$ and $\rm NH_3 + N_2$ atmospheres

Comparison of consumption of $\rm NH_3$ and $\rm NH_3 + N_2$ atmospheres is much more difficult than the comparison of consumtion of $\rm NH_3$ and $\rm NH_3 + NH_3$ diss. ones. In a process using the $\rm NH_3 + N_2$ mix, the net flux of nitrogen atoms from the gas phase to the surface, $\rm J_N^{g}$, is controlled by the following parameters: composition of the atmosphere in the retort, its nitriding potential $\rm K_N$, kinetic parameters (such as diffusion coefficients). The relationship applies:

$$J_N^g = \beta \cdot (c_N^g - c_N^s) \tag{1a}$$

where β is the transfer coefficient from the gas to the solid phase (also called the reaction constant), and is expressed as:

$$\beta = \beta_0 \cdot p_{H_2} \cdot \exp\left(-Q_1 / RT\right) \tag{1b}$$

 β_0 – experimental constant in m/s/atm, pH $_2$ – partial pressure of hydrogen, Q – activation energy, R – gas constant, T – absolute temperature, $c_{\rm N}{}^{\rm g}$ – surface concentration of nitrogen in equilibrium with the atmosphere, which is a function of the nitriding potential of the atmosphere, $c_{\rm N}{}^{\rm s}$ – actual nitrogen concentration at the metal's surface. Using Fick's first law it follows:

$$\beta \cdot (c_N^g - c_N^s) = D_M^N \frac{dc}{dx}$$
(2)

 $D_{\ M}^{N}$ – diffusion coefficient of nitrogen in the superficial layer on steel; ${}^{dc/}_{dx}$ – concentration gradient at the surface of the superficial layer.

Diluting NH₃ with molecular nitrogen lowers the partial pressure of hydrogen pH₂, and, in consequence (according to 1b) diminishes the transfer coefficient β of nitrogen. Apart from this, the presence of N₂ has two counter-acting effects on the potential in the retort and concurrently on the equilibrium concentration of nitrogen c_N^s. On the one hand, a simultaneous lowering of the NH₃ and H₂ partial pressures increases the potential K_N. On the other hand, according to Le Chatelier's rule, it promotes the dissociation of NH₃, decreasing K_N. The resulting effect of both tendencies depends on the surface area of

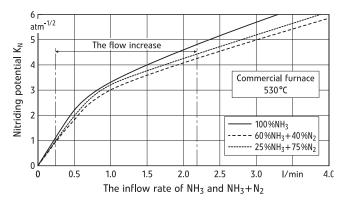


Fig. 7. Effect of the inflow rate of NH_3 and $NH_3 + N_2$ atmospheres and nitrogen content in the $NH_3 + N_2$ atmosphere on the nitriding potential in the retort. Industrial nitriding pit-type furnace, 1 m³ working chamber

the retort and of the charge, temperature, atmosphere flow, and $\rm N_2$ content. In typical industrial retorts, these tendencies are approximately balanced, which is shown in Fig. 7.

This finally leads to the conclusion that diluting of ammonia reduces the flux J_N^{g} . Accordingly, in order to maintain the same flux J_N^{g} in the $NH_3 + N_2$ as in an atmosphere of NH_3 , it is necessary to increase the equilibrium concentration of nitrogen c_N^{g} , or nitriding potential K_N . This is shown schematically in Fig. 7, where a horizontal arrow indicates the degree of dilution of ammonia with nitrogen, and a vertical arrow the required increase of potential K_N . For a more accurate calculation of the effect of N_2 content on the consumption of the $NH_3 + N_2$ atmosphere, it would be necessary in the determination of the flux J_N^{g} , to employ the relationship (2).

However, in view of the lack of physico-chemical data required for this approach (diffusion coefficients of nitrogen in the different phases of the nitrided layer, the transfer coefficients β of nitrogen from the gas phase to the metallic phases) this is not possible.

The analysis of many processes carried out in industrial furnaces has proven [20] that replacing $\rm NH_3$ with a $\rm NH_3 + N_2$ mix leads to an increase in the consumption of $\rm NH_3$ up to several times over, depending on the $\rm N_2$ content. Concurrently, the additional consumption of nitrogen has to be considered. In consequence, a process utilizing a $\rm NH_3 + N_2$ atmosphere has to be more expensive than that run with $\rm NH_3$ only. However, consumption of the $\rm NH_3 + N_2$ mix is still much lower than that of the $\rm NH_3 + NH_3$ diss. atmosphere. In the latter case, mentioned above, the cost of a dissociator has to be added to the high price of the atmosphere.

6 Other benefits and limitations of nitriding in 100 % NH₃

A process which utilizes only NH₃, brings, in comparison with processes making use of two-component mixes, other substantial benefits, both practical and economical. Because the kinetics of the layer growth in the case of the NH₃ + N₂ atmosphere is controlled by two parameters (nitriding potential K_N and the N₂ content), the design of processes requires many more experimental trials than in the case of atmospheres of NH₃ or NH₃ + NH₃diss., in which the kinetics is controlled by one parameter only (K_N). For this reason, the precision of predicting as well as the reproducibility of results is higher for processes in NH₃ (also when using NH₃ + NH₃diss.) than when using NH₃ + N₂. These additional economical factors argue in favor of processes in 100 % NH₃.

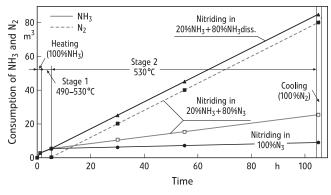


Fig. 8. Comparison of consumption of $NH_3 + N_2$, $NH_3 + NH_3$ diss. and NH_3 atmospheres in the total nitriding process. Industrial nitriding pit-type furnace, 1 m³ working chamber

A process with only NH_3 requires only one gas flow controller, while for processes with $NH_3 + N_2$ and $NH_3 + NH_3$ diss. atmospheres, two such controllers have to be used.

For precise regulation and control of the chemical composition of atmosphere obtained from NH_3 (also from $NH_3 + NH_3$ diss.), it is sufficient to determine one gas component in the retort, whereas in case of the $NH_3 + N_2$ atmosphere two are necessary. Also from this point of view, the process in NH_3 is simpler and less expensive than that using $NH_3 + N_2$. The NH_3 + NH_3 diss. atmosphere not only requires two gas flow controllers and the determination of one component in the atmosphere but also, as mentioned above, a separate dissociator, which adds to the cost of the equipment.

A process using only $\rm NH_3$ (also $\rm NH_3 + \rm NH_3$ diss.) may be designed, and even controlled with the aid of existing models of kinetics of growth [15-18, 21]. The $\rm NH_3 + \rm N_2$ atmosphere requires the development of more involved models and additional physico-chemical data (transfer coefficients of nitrogen to ε and γ' phases at the surface of iron and steel workpieces). Unfortunately, because of the complex mechanisms of processes in the $\rm NH_3 + \rm N_2$ atmosphere, and due to the lack of experimental data, the development of such models is as yet not feasible.

Savings in the consumption of gases, indicated above, concern only the second stage of the nitriding process, in which the supply of nitrogen atoms from the gas phase to the metal surface has to be restricted. There is no opportunity for savings in the first stage, during which, in view of the required intensive supply of atomic nitrogen to the surface, a high flow of NH_3 is necessary. Also, this does not apply to the consumption of gases (NH_3) during heating up as well as during the cooling period (N_2). Thus, to estimate realistic savings in gas consumption, attributable to the new method, it is necessary to consider total gas consumption in all process stages.

The corresponding results are presented in graphical form in Fig. 8. It has to be recognized that the benefits of using 100 % $\rm NH_3$ for nitriding increase with the process (stage 2) time. Thus, it is not reasonable to seek savings in short-term processes (such as those applied to non-alloyed steel), which are performed in 100 % ammonia at high inflow rate.

7 Conclusions

The results of the investigation indicate that by performing the process using NH_3 alone it is possible to produce nitrided layers identical to those obtained in two-component atmos-

pheres NH₃ + NH₃diss. and NH₃ + N₂. In such a process it is possible to control the composition of the atmosphere through the regulation of NH₂ flow into the retort, with occasional closing of the flow, thereby obtaining a precision of control similar to that available with the NH₃ + NH₃diss. atmosphere.

The process using only NH₃ brings the benefit of low consumption of gas and can be realized with a simpler and, by the same token, less expensive control system, compared with processes using two-component atmospheres NH₂ + NH₂diss. and $NH_2 + N_2$.

The process using NH₂ alone is similar to that using NH₂ + NH₂diss. in terms of easy control of the kinetics of growth of the nitrided layer. Easier, in fact, than in the case of the atmosphere $NH_3 + N_2$.

The authors acknowledge the support received from Nitrex Metal Inc. of Montreal, Canada. We also wish to thank Dr. George Tymowski for constructive help in the investigation and preparation of this paper.

References

- 1. Patent US 1,487,554 (1924) 2. *Knerr, C. H.; Rose, Th. C.; Filkowski, J. H.:* Gas Nitriding. ASM Hand-
- Knerr, C. H.; Rose, In. C.; Fukowski, J. H.: Gas Mitriding. ASM Handbook TM, Vol. 4, Heat Treating, 1991, p. 387–425
 Sinha, A. K. (Ed.): Physical Metallurgy Handbook. McGraw-Hill, New York, 2003, p. 16.89. ISBN 0-07-057986-5
 Floe, C. F.: A Study of the Nitriding Process. Transactions for American Society for Metals 32 (1944), p 134–149
 Bever, M.; Floe, C. F.: Case hardening of steel by nitriding. In: Surface Distribution and Comparison II. C. Automatican Society for Metals 32 (1944), p 134–149
- Protection Against Wear and Corrosion, H. S. Avery (Ed.). ASM, 1953, p. 123-143
- Minkevich, A. N: Thermochemical treatment of metals and alloys. 6.
- Mashinostroenie (1965), p. 331 Sorokin, Yu V.; Minkevich, A. N.: Nitriding steel in a mixture of nitrogen and ammonia. MiTOM (1966) 5, p. 49–52 7.
- Patent PL85924B (1976)
- Haś, Z.; Kula, P.: Nitrovac'79 Nowa technologia obróbki cieplnochemicznej elementów maszyn i narzędzi. Inżynieria Materiałowa
- Małdzinski, L.: Thermodynamic, kinetic and technological aspects of the producing nitrided layers on iron and steel in processes of gas nitriding (in Polish). Habilitation thesis, Technical University Poznan, Poland, Wydawnictwo Politechniki Pozna skiej, 2002. -JSDN 02 74425 544.0 ISBN 83-7143-564-9
- Przyłęcki, Z.; Małdzinski, L.: Investigation of equilibrium conditions in the system NH₃-H₂-N₂-nitride ε. Proc. III. Int. Conf. Carbides, Nitrides, Borides, Poznań-Kołobrzeg/Poland, 1984, p. 116–129.

- 12. *Przylęcki, Z.; Małdzinski, L.:* Conditions of equilibrium between NH_3-H_2 gas mixture and nitrogen in γ ' nitride. Proc. IV. Int. Conf. Carbides, Nitrides, Borides, Poznań-Kołobrzeg/Poland, 1987, p. 153-162
- 13. Małdzinski, L.; Przyłęcki, Z.; Kunze, J.: Equilibrium between ammonia-hydrogen mixtures and the ε-phase of iron. Steel Research 57
- (1986) 12, p. 645-649
 14. Somers, M. A. J.; Kooi, B. J.; Ma∏dzi ski, L.; Mittemeijer, E. J.; van der Horst, A. A.; van der Kraan, A. M.; van der Pers, N. M.: Thermodynamics and long-range order of interstitial in an h. c. p. lattice; nitrogen in e-Fe₂N_{1,2}. Acta Metal. Mater. 45 (1997), p. 2013–2025 15. *Mittemeijer, E. I.; Somers, M. A. J.:* Thermodynamics, Kinetics, and
- Process Control of Nitriding. Surface Eng. 13 (1997) 6, p. 483–497
 16. Mittemeijer, E. J.; Somers, M. A. J.: Thermodynamics, Kinetics, and Process Control of Nitriding. Proc. 10th Congr. of IFHT, 1.-5.9.99,
- Brighton/UK, T. Bell, E. J. Mittemeijer (Ed.), p. 83–117
 17. Torchane, L.; Bilger, P.; Dulcy, J.; Gantois, M.: Control of Iron Nitride Layers Growth Kinetics in the Binary Fe-N System. Metall. Mater.
- Layers Growth Knieucs in the Briary Fe-IN System. Metall: Mater. Trans. A, 27A (1996) July, p. 1824–1834
 18. Somers, M. A. J.; Ma∏dzi ski, L.; Sommer, T.: The Growth of the ε+γ' nitrides on pure iron. In preparation
 19. Małdzinski, L.; Liliental, W.; Tymowski, G.: Causes of Inhomogeneity of Chemical Composition of Nitriding Atmospheres in the Furnace Retort. Proc. 19th ASM, 1-4 November, 1999, Cincinnati, Ohio, USA, S. J. Midag (Tel.) p. 170–196 S. J. Midea (Ed.), p. 178-186
- 20. Tacikowski, J.: unpublished report, 1998
- Małdzinski, L.; Liliental, W.; Tymowski, G.; Tacikowski, J.: New Possibilities of Controlling the Gas Nitriding Process by Utilizing Simulation of Growth Kinetics of Nitride Layer. Proc. 12th Int. Conf. on Surface Modification Technologies, 12-14 October, 1998, ASM Rosemont, Illinois, USA, p. 215-226 and Surface Eng. 15 (1999) 5, p. 377-384

The authors of the paper

Prof. Dr. hab. Inz. Leszek Maldzinski, born in • ??? • , is Professor at the University of Technology, Poznan/Poland and scientific adviser of Nitrex Metal Ínc. Canada.

Prof. Dr. Inz. Jan Tacikowski, born in · ??? · , is Professor at the Institute of Precision Mechanics, Warsaw/Poland.

Manuscript submitted in November 2006.

Dieser Beitrag hat die Dokumentennummer HT 100 ••• und steht unter www.HTM-Journal.de für Sie zum Download bereit.

Concept of an economical and ecological process of gas nitriding of steel

L. Małdziński, J. Tacikowski HTM 61 (2006) 6, p. 000–000, 8 figs., 21 literature references

An economical, ecological version of the gas nitriding process is presented. It allows a substantial reduction in the consumption of industrial gases compared with currently used processes. Also it leads to a simplification of nitriding installation as well as the process itself, while allowing maintenance of full control over the kinetics of the layer growth. The concept was developed from studies of the thermodynamics and kinetics of growth of nitrided layers on iron and steel. It has been verified in trials under industrial conditions.

Konzept eines ökonomischen und ökologischen Prozesses des Nitrierens von Stahl mit Gas

L. Małdziński, J. Tacikowski HTM 61 (2006) 6, S. 000–000, 8 Bilder, 21 Literaturangaben

Es wird eine ökonomische, ökologische Möglichkeit des Nitrierprozesses mit Gas vorgestellt.